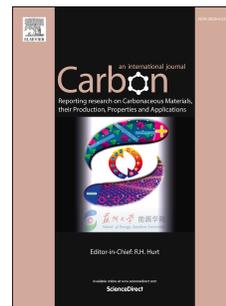


Journal Pre-proof

One-pot preparation of zwitterionic graphene nanosheets with exceptional redispersibility and its application in pickering emulsions

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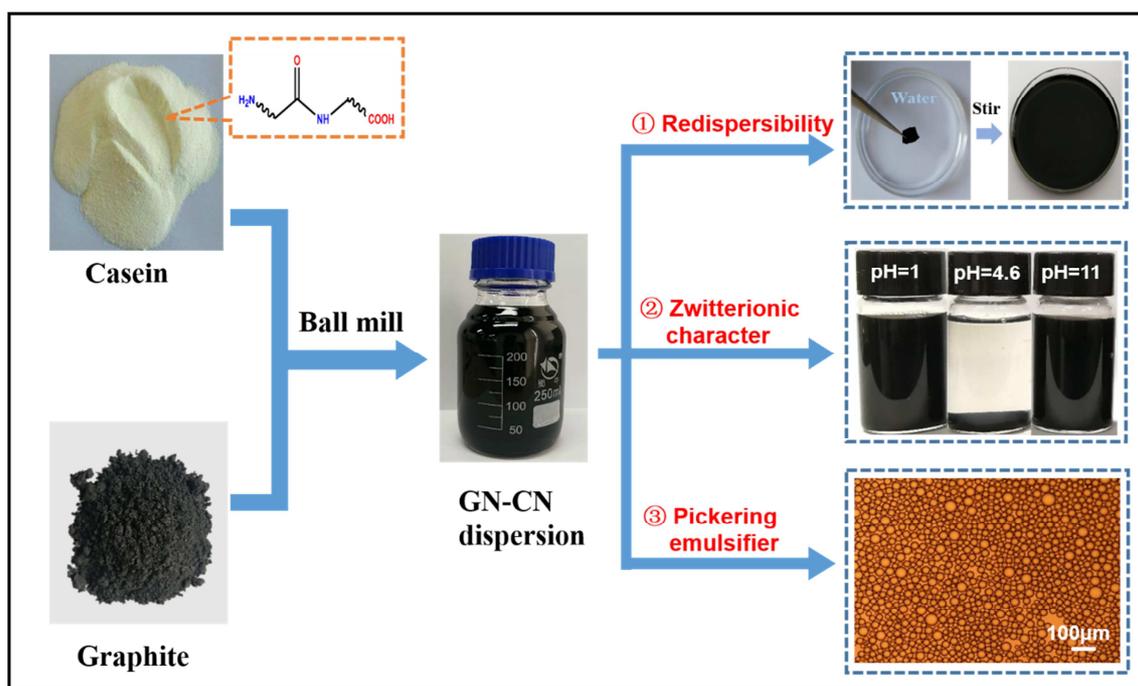
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The facile preparation of zwitterionic graphene nanosheets with exceptional redispersibility and its application as Pickering emulsifier have been demonstrated.

24 **Abstract**

25 Using biomass materials as exfoliating agents to produce graphene is a promising strategy due to
26 their biocompatibility, reproducibility and sustainability. However, it's still a challenge to prepare
27 high-concentration graphene dispersion using biomass exfoliating agents. Here, we present a green
28 and facile method to prepare few layered graphene nanosheets in aqueous solution by one-pot ball
29 milling in the presence of protein, casein (CN). The results show that CN protein could be acted as
30 an efficient exfoliating and stabilizing agent, producing water-soluble graphene nanosheets with a
31 concentration up to 2.14 mg ml^{-1} . As CN protein contains both amino and carboxyl groups, thus the
32 prepared graphene nanosheets decorated with CN protein (GN-CN) show zwitterionically charged
33 character and thereby can be stably dispersed under both acidic and alkaline conditions. Moreover,
34 the oven-dried GN-CN hybrid exhibits exceptional redispersibility in water with a concentration as
35 high as 100 mg ml^{-1} and even more. Benefiting from the zwitterionic character, GN-CN shows the
36 Pickering emulsion effect over a wide pH range. As a proof of concept, the application of the
37 zwitterionic GN-CN as a Pickering emulsifier to prepare microcapsule phase change composite
38 material has been successfully demonstrated.

39

40 **1. Introduction**

41 As a result of the unparalleled physical properties and promising applications, graphene has
42 attracted tremendous attention since its discovery in 2004^{1,2}. Although the unique two-dimensional
43 (2D) structure gives it excellent mechanical strength ($\sim 1 \text{ TPa}$), superior electrical (6000 S cm^{-1}) and
44 thermal conductivity ($\sim 5300 \text{ W m}^{-1} \text{ K}^{-1}$)³⁻⁶, there are still many challenges to produce processable
45 graphene nanosheets in large quantities and achieve suitable applications. Typically, liquid-phase

46 exfoliation by shear force or sonication is considered to more suitable for preparing high-quality and
47 low-cost graphene on a large scale⁷⁻¹⁰. According to previous studies, suitable organic solvents^{7, 11, 12}
48 and some water dispersible stabilizers^{1, 13-15} can serve as exfoliation media for exfoliating and
49 stabilizing graphene nanosheets. Compared with organic solvents, stabilizer-assisted exfoliation in
50 aqueous media is a green and promising way to produce highly dispersible graphene nanosheets. The
51 adsorbed stabilizers on graphene surface prevent their stacking, making it a more stable dispersion.
52 Moreover, noncovalent functionalization between stabilizers and graphene sheets endows graphene
53 interesting properties, which can broaden the applications of graphene¹⁵⁻¹⁸. However, it should be
54 pointed out that many stabilizers are synthesized by chemical method, which are toxic,
55 environmentally unfriendly and non-renewable, and run counter to the theme of sustainable
56 development^{15, 18, 19}.

57 Recently, some natural materials, such as gum arabic¹⁷ and cellulose nanocrystal²⁰ have been used
58 as intriguing liquid-phase exfoliating agents due to their biocompatibility, renewability and
59 non-toxicity^{16, 17, 20-25}. The resultant graphene nanosheets decorated with biomacromolecules via the
60 covalent or non-covalent interaction exhibit remarkable stability, high yield and other interesting
61 performance, attributed to synergetic effects between graphene and the corresponding
62 biomacromolecules. As a typical biomacromolecule, protein exists widely in biology body of nature,
63 such as plants, animals and microorganisms. A few studies demonstrated that some specific proteins
64 are effective exfoliating agents for graphene and other two-dimensional materials in aqueous
65 solutions²³⁻²⁵. As an example, Guan et al.²³ demonstrated bovine serum albumin can be used to
66 exfoliate and stabilize graphene and other two-dimensional nanomaterials. In another work, Ge et
67 al.²⁴ produced gelatin-adsorption graphene by sonication in gelatin solutions. The resultant graphene

68 significantly improves its compatibility with the gelatin substrate and the mechanical properties of
69 their composites. However, the yield and concentration of graphene prepared with the two proteins
70 are relatively low, so it is promising to explore a new protein to prepare high-concentration graphene
71 dispersion with high efficiency.

72 Casein (CN) is a globular amphoteric protein with an isoelectric point (IP) of pH 4.6, and it is
73 readily available from milk^{26,27}. Since the molecule contains both amino and carboxyl groups, CN
74 can be dispersed in both acidic and alkaline conditions, which makes casein a zwitterionic protein.
75 Meanwhile, CN exhibits many specific properties such as excellent emulsification, stability and
76 adhesion due to its distinct amphiphilic structure. Therefore, we speculated that such a protein with
77 both zwitterionic and amphiphilic structure might be used to exfoliate graphite into graphene
78 nanosheets in the aqueous solution and endow the resultant graphene nanosheets zwitterionic
79 character.

80 Herein, we presented the feasibility of CN-assisted exfoliation of graphene nanosheet by the
81 one-pot ball milling method. Aqueous dispersions of few layered graphene nanosheets at high
82 concentration have been successfully prepared. To one's surprise, the resultant GN-CN hybrid
83 exhibits an extraordinary redispersibility in water with a high concentration (100 mg ml⁻¹). Moreover,
84 the GN-CN is zwitterionic and shows doubly pH-responsive behavior due to the distinct chemical
85 structure of CN protein. Finally, GN-CN was used to stabilize Pickering emulsion over a wide pH
86 range and its application in the preparation of microcapsule phase change material was studied.

87 **2. Experimental Section**

88 2.1. Materials

89 Natural graphite (325 mesh, 99.95% purity) was purchased from Qingdao carbon Co. Ltd. CN

90 powder was purchased from Hua 'an Biological Products Co. LTD (Ningxia, China). Toluene was
91 purchased from Chemical Industry Co. LTD (Shandong, China). CCK-8 (Cell Counting Kit-8) was
92 purchased from Beyotime Biotechnology Co. Ltd (Shanghai, China). Calcien AM and propidium
93 iodide (PI) were obtained from Solarbio life science Co. Ltd (Beijing, China). Stearic acid (SA) was
94 obtained from Beilian Fine Chemicals Development Co. LTD (Tianjin, China).

95 2.2. Preparation of CN dispersion

96 CN dispersion was prepared by dissolving the CN powder in an alkaline solution. Typically, CN
97 powder (10 g) was suspended in deionized water (85 g), and stirred at 400 rpm at 50 °C for 4 h. Then
98 25% ammonia (5 g) was added into the CN dispersion. After continuous stirring for 1 h to dissolve
99 the CN, the solution was cooled down to room temperature to form a 10% CN aqueous dispersion.
100 Prior to use, it was diluted to a designated concentrations using deionized water.

101 2.3. Preparation of graphene nanosheets decorated with CN (GN-CN)

102 Natural graphite powder (3 g) was first added in CN aqueous dispersions with different concentration
103 (300 ml, 0.25 mg ml⁻¹-9 mg ml⁻¹). Then the mixtures were ball milled at 185 rpm for certain time
104 (5-40 h) through a planetary ball mill machine (MITR-YXQM-2L) with the stainless steel grinding
105 chamber having a volume of 0.5 L and four sizes of stainless steel mill balls (2 mm, 5 mm, 8 mm and
106 10 mm in diameter). Unexfoliated graphite was removed firstly by centrifugation at 4000 rpm for 30
107 min to remove the bulk graphite, and then was centrifuged at 8000 rpm for 30 min to remove the
108 thick graphene sheets.

109 2.4. Preparation of wide pH-responsive Pickering emulsion

110 The Pickering emulsions were prepared by toluene (6 ml) and GN-CN dispersions (9 mg ml⁻¹, 3 ml)
111 with different pH values. After mixing, the mixtures were emulsified with a homogenizer at 6000

112 rpm for 5 min at room temperature (RT). The pH of the GN-CN dispersions were adjusted by adding
113 0.5 M HCl or NaOH, and the resultant Pickering emulsions were observed and characterized after
114 standing for 24 h at RT.

115 2.5. Cytotoxicity evaluation of GN-CN dispersion

116 CCK-8 assays were performed to evaluate the cytotoxicity of GN-CN dispersion. The procedure was
117 as follows: MLg cells were incubated in DMEM-12 culture medium for 24 h and then incubated in
118 GN-CN dispersions with different concentrations. After incubation for 24 h, the relative cell viability
119 was checked by the CCK-8 assay. For live/dead cell staining, cell viability was tested by
120 double-staining procedure using Calcein AM and propidium iodide (PI). In brief, after incubation for
121 24 h, cells were incubated with Calcein AM for 30 min at 37 °C. And then, cells were incubated with
122 PI for 5 min at room temperature. After incubation, the samples were washed with phosphate buffer
123 saline (PBS) twice and photographed using a Leica TCS SP5 confocal microscope. Cell viability was
124 calculated via the staining images, in which the green and red fluorescence corresponded to the
125 viable cells staining with Calcein AM and the dead cells staining with PI, respectively.

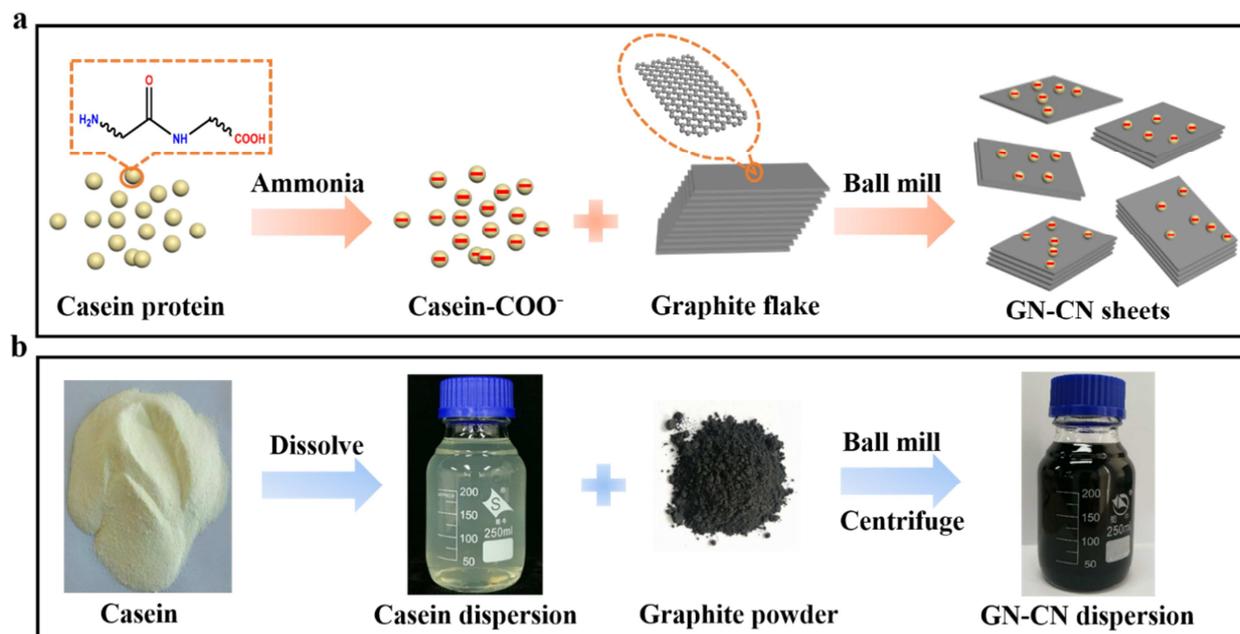
126 2.6. Preparation of composite microcapsule phase change materials (mPCMs)

127 The mPCMs was prepared by a Pickering emulsion method. Typically, SA (1 g) was added to a series
128 of GN-CN dispersions (9 mg ml^{-1} , 10 ml) with different pH values, followed by heating the mixture
129 to 90 °C in an oil bath to allow the melting of SA. Then, the mixture was emulsified with a
130 homogenizer at 7000 rpm for 10 min to form an emulsion of SA microcapsule phase change
131 materials, which was encapsulated by GN-CN. The resulting emulsions were allowed to cool
132 naturally at RT for 24 h. To obtain the solid state mPCMs, the pH of emulsions were adjusted to near

133 the isoelectric point of CN to allow SA droplets encapsulated to settle. As-prepared solid mPCMs
134 was dried at 50 °C for a night.

135 2.7. Characterization

136 UV-vis absorption were recorded using a UV-vis spectrophotometer (Shimadzu UV-2550) in the
137 scanning range from 200 nm-800 nm. X-ray diffraction (XRD) of graphene nanosheets were
138 characterized by an X-ray diffractometer (RIGAKU Ultima IV) using Cu K α radiation ($\lambda = 0.15406$
139 nm). Raman measurements were performed with a Raman Spectrometer (InVia Series Laser
140 Confocal Micro, RENISHAW) using a 532 nm excitation laser beam. Atomic force microscopy
141 (AFM) images were taken using a Dimension Icon (BRUKER) with PeakForce Tapping mode.
142 Transmission Electron microscopy (TEM) analyses were performed using a Tecnai F20 (FEI).
143 Thermogravimetric analysis (TGA) was carried out using a Perkin Elmer TGA 6 (Perkin Elmer) with
144 a heating rate of 10 °C/min ranging from 30 to 800 °C under nitrogen atmosphere. The Zeta Potential
145 measurements were measured by a Dynamic Light Scattering (NanoBrook Omni) at 25 °C. The CD
146 spectra were measured using a J-1500 CD Spectrometer (JASCO). The fluorescence images of
147 cytotoxicity were photographed using a Leica TCS SP5 confocal microscope (Leica Microsystems,
148 Germany). The optical micrographs of the prepared emulsions on transparent glass slides were taken
149 using an ECLIPSE LV100N POL (Nikon). Differential scanning calorimetry (DSC) was tested using
150 a Q20 (TA Instrument) at a heating and cooling rate of 20 °C/min. The morphology of phase change
151 materials was observed using a Scanning Electron Microscope (JEOL SEM 6700).

152 **3. Results and discussion**

153 **Figure 1.** Schematic illustration of GN-CN preparation process. (a) Mechanism of exfoliation and
 154 stabilization of GN-CN sheets. (b) Ball milling exfoliation process of GN-CN in CN dispersion.

155

156 **3.1. Preparation and Characterization of GN-CN dispersion**

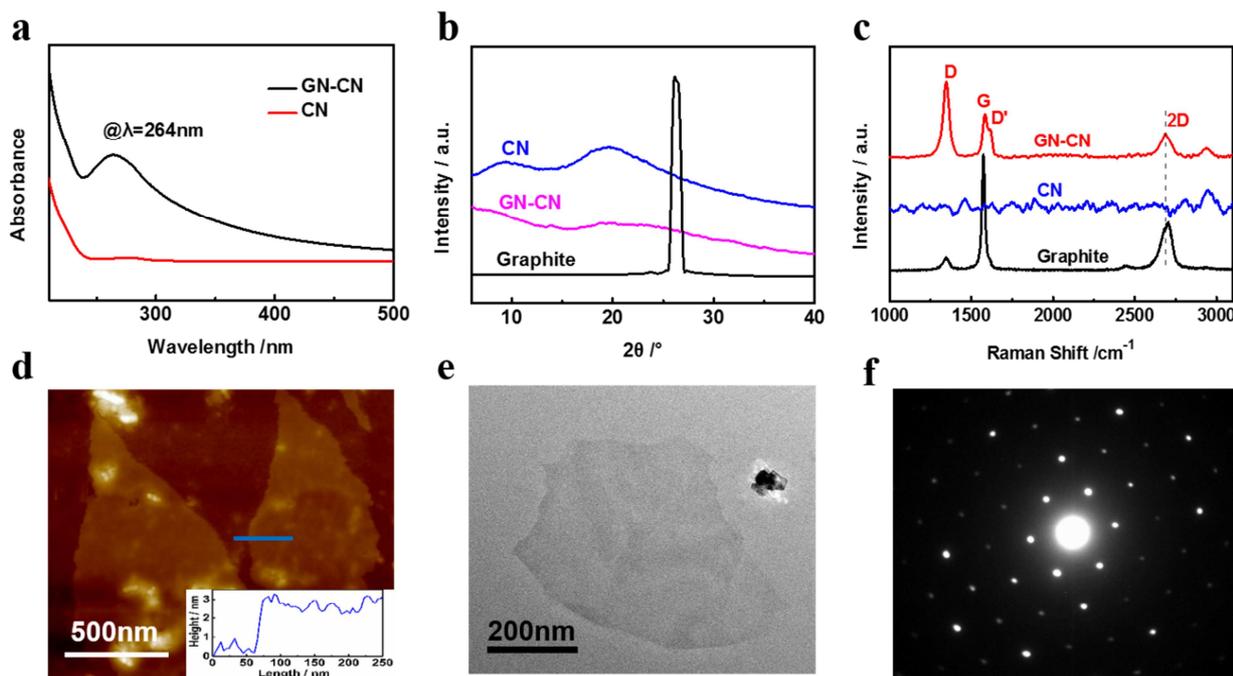
157 The preparation process is illustrated in Fig. 1. First, globular amphoteric CN was dissolved in
 158 deionized water by adding a certain mass of ammonia to obtain anionic charged CN dispersions
 159 (COO^-) with high dispersion stability. Afterwards, the mixture of CN dispersion (COO^-) and
 160 graphite was subjected to liquid-phase ball milling. Owing to the hydrophobic interaction between
 161 the CN molecular chains and the graphite sheets, the CN molecules were tightly adsorbed on the
 162 surface of the graphene sheets, resulting in the anionic charged graphene dispersion after ball milling.
 163 It's worth noting that as the pH value changes, the graphene sheets become positively charged due to
 164 the protonation of the amino group in the CN molecule, which will be confirmed in later.

165 Preparation conditions have a significant influence on the concentration of prepared graphene

166 dispersions and exfoliation efficiency according to previous study²⁰. Here, we studied the effects of
167 ball-milling time, CN concentration and pH of CN dispersion on graphene concentration and
168 exfoliation efficiency. UV–vis spectroscopy was employed to evaluate the exfoliation efficiency. As
169 shown in Fig. S1a, b, the absorption intensity at 264 nm increases with ball milling time within 40 h,
170 indicating the increasing yield of graphene. However, longer ball milling time will damage the size
171 and quality of the exfoliated graphene according to our previous research¹⁶. In addition, higher CN
172 concentration will result in higher graphene yield within a certain concentration range of CN (Fig.
173 S1c, d). The proportion of graphene in the prepared GN-CN was estimated from TGA profile (Fig.
174 S2a). Fig. S2b shows that both graphene concentration and yield increase with the CN concentration
175 but at the expense of decreasing graphene/GN-CN concentration ratio in the GN-CN dispersion. The
176 highest graphene concentration and yield are 2.14 mg ml^{-1} and 21.4% respectively (Table S1), giving
177 CN a distinct advantage over other natural materials which we will discuss in the following section.
178 We predict that too high concentration of CN will reduce the concentration and yield of graphene due
179 to the increasing viscosity and CN aggregation. Consider the effect of ball milling time and CN
180 concentration comprehensively, the concentration of GN-CN dispersion used in our experiments and
181 characterizations is both 1 mg ml^{-1} with 30 h ball-milling except for special instructions. In addition,
182 the dependence of absorption intensity at 264 nm on pH of CN dispersion suggests that GN-CN
183 dispersion can be prepared successfully only under strong acid and alkaline conditions in Fig. S3.
184 That may be attributed to the better protonation of the amino and carboxyl groups facilitate the
185 exfoliation and stabilization of GN-CN.

186 The UV–vis absorption spectrum for GN-CN dispersion was employed to study the conjugated
187 structure of the exfoliated graphene nanosheets. As shown in Fig. 2a, compare with the negligible

188 absorption of CN, the UV-vis spectrum of GN-CN exhibits a strong absorption peak around 264 nm,
189 which is ascribed to the π - π^* transitions of aromatic C=C bonds²⁰. The result demonstrates the
190 conjugated structure of GN-CN nanosheets. The XRD pattern of the GN-CN nanosheets didn't show
191 the reflection peaks at 26.5° from bulk graphite²⁸ in Fig. 2b, indicating that there is no stacking of
192 layers, i.e., graphite was exfoliated into graphene nanosheets successfully. It's worth noting that the
193 wide flat peak of GN-CN over a wide range (15-30°) may be attributed to the amorphous diffraction
194 of CN. From the Raman spectrum of graphite in Fig. 2c, we observed that graphite exhibits its
195 characteristic spectrum with a D band at 1348 cm⁻¹, G band at 1576 cm⁻¹ and 2D band at 2706 cm⁻¹²⁰.
196 The strong D band and D' band at 1622 cm⁻¹ in the Raman spectrum of GN-CN suggest the defects
197 introduced during ball milling. Previous studies had proposed that the defects introduced in the
198 process of mechanical exfoliation are mainly at the edges of graphene rather than the basal plane^{9,11}
199 ^{25, 29, 30}. The layer number of graphene sheets was also estimated by analyzing Raman spectra.
200 Compared with the graphite, the 2D band of GN-CN downshifts to lower frequency (15cm⁻¹),
201 demonstrating the few layerd graphene nanosheets in the GN-CN dispersions³¹. Furthermore, related
202 research proposed that the ratio of I_{2D}/I_G can be used to evaluate the number of graphene layers. It is
203 calculated that the I_{2D}/I_G value is 0.59 for the GN-CN, corresponding to approximately 5 layers or
204 less considering the aggregate of graphene sheets³².



205

206 **Figure 2.** (a) UV-vis spectra of CN and GN-CN dispersion. (b) X-ray diffraction patterns of starting
 207 graphite, CN and GN-CN. (c) Raman spectra of GN-CN, CN and graphite. (d) AFM image of
 208 GN-CN with CN protein on its surface. (e) Typical TEM image of few layered GN-CN. (f)
 209 Selected-area electron diffraction (SAED) pattern of GN-CN.

210

211 The thickness and morphology of as-prepared GN-CN were further characterized by atomic force
 212 microscopy (AFM). Fig. 2d shows the thickness of a representative GN-CN from the as-prepared
 213 dispersion is about 2.05 nm, corresponding to 2–3 layer graphene sheets^{1, 20, 24}. TEM image also
 214 provides evidence of few layered graphene exfoliation as shown in Fig. 2e. The selected area
 215 electron diffraction pattern (SAED) in Fig. 2f shows a typical hexagonal symmetry structure,
 216 confirming the high crystallinity of the GN-CN³³. To find out more about the morphology of
 217 graphene and show statistical distributions of size and thickness of graphene, the GN-CN dispersion
 218 was further centrifuged at 12500 rpm for 30 min, and then the obtained sediment was re-dispersed in

219 deionized water of pH=10. Several other typical graphene nanosheets with lateral size in the range of
220 tens of nanometers (Fig. S4) to tens of micrometers (Fig. S5) were also identified. The lateral size
221 and thickness of 80 different graphene sheets were examined and the statistical distribution
222 histograms were shown in Fig. S6. The large sized graphene sheets (lateral size larger than 2 μm)
223 accounts for only 5% while the graphene sheets with lateral size within 400 nm accounts for 84%.
224 Meanwhile, the thickness of graphene sheets is mainly distributed below 8 nm, accounting for 88%.
225 These results indicate that the as-prepared GN-CN is mainly of small size and large sized graphene
226 sheets are less abundant and can be negligible.

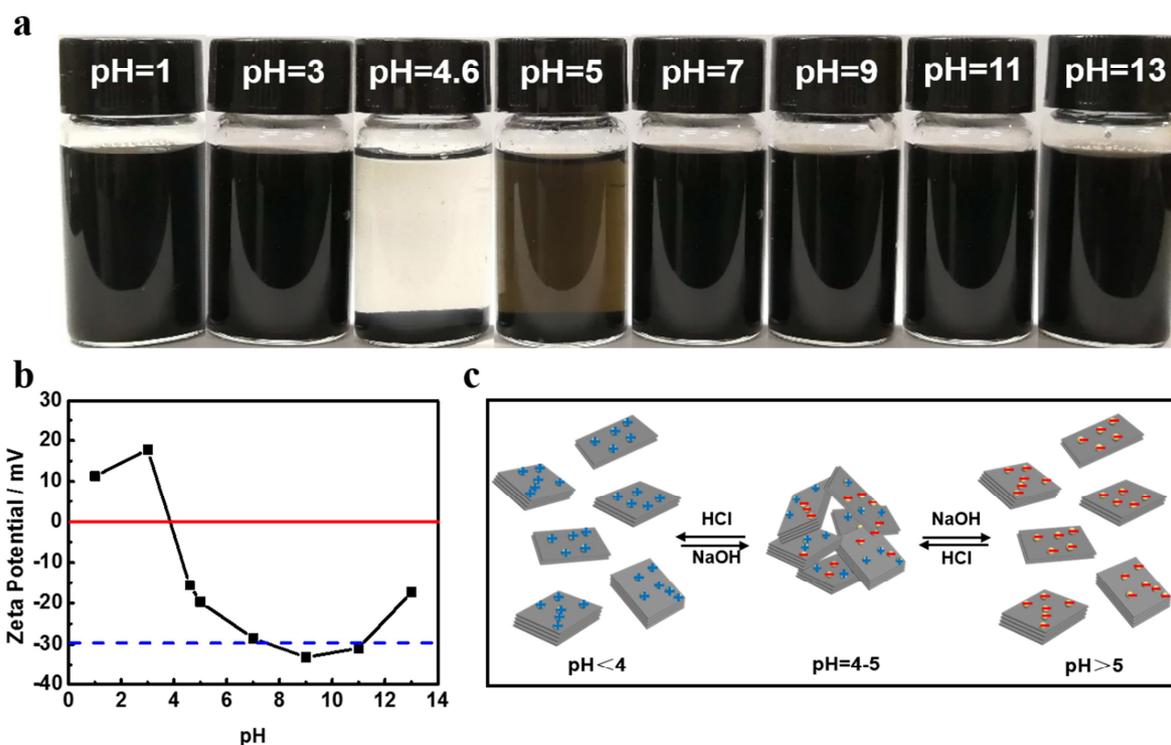
227 To characterize the stability of GN-CN dispersion, we measured the zeta potential of the GN-CN
228 dispersions prepared with different ball milling time and CN dispersion concentration by dynamic
229 light scattering. As shown in Fig. S7, almost all the GN-CN dispersions exhibit zeta potentials below
230 -30 mV, indicating a certain amount of CN absorbed on the graphene nanosheets and formed a stable
231 dispersion. The CD spectra of original CN dispersion and GN-CN dispersion were measured to
232 determine whether the structure of CN was affected after ball milling. The CD spectra in Fig. S8
233 show that CN and GN-CN dispersions prepared with different CN concentration both exhibit
234 negative absorption peaks at 201 nm, suggesting that the secondary structure of CN was not damaged
235 and the CN remained stable. As a result, we conclude that CN is an efficient exfoliating and
236 stabilizing agent for preparing few layered graphene nanosheets.

237 To compare CN with other nature materials used as exfoliating agents, the related information of
238 graphene exfoliated were summarized in Table S2. Graphene concentration and yield are very
239 important evaluation indicators for liquid-phase exfoliation of graphite. The results show that the
240 concentration and yield of graphene using CN as exfoliating agent are higher than most of other

241 natural materials and all proteins (gelatin, HFBI and bovine serum albumin) listed in Table S2. In
242 addition, we find it difficult to achieve the exfoliation of single layered graphene by using natural
243 materials and the few, multilayers graphene and thick “graphite flake” are also a ubiquitous trend.
244 Therefore, high concentration and yield GN-CN with different lateral size (from tens of nanometers
245 to tens of micrometers) and thickness (few to multi layers) was successfully prepared.

246 The exfoliation mechanism of GN-CN can be explained by binding energy of CN on graphene
247 layers. CN is an amphiphilic protein with distinct hydrophilic and hydrophobic regions, and will
248 result in a binding energy due to the hydrophobic interaction between the hydrophobic part of CN
249 and graphene layers. According to previous reports²³, the binding energies of different groups of
250 protein on graphene vary greatly. The peptide bonds in the amino acid sequence of the protein will
251 generate a strong binding energy on the graphene sheets, which will make the peptide bond tightly
252 binding on the graphene surface, while the hydrophilic polar groups are exposed to the water. On the
253 contrary, the nonpolar benzene rings in phenylalanine exhibit an unstable binding on the surface of
254 graphene due to the weak binding energy. CN is a polymer consists of four types of CN monomers,
255 α_{s1} -casein (α_{s1} -CN), α_{s2} -casein (α_{s2} -CN), β -casein (β -CN) and κ -casein (κ -CN) and each CN
256 monomer has its own unique amino acids sequence³⁴. The types and numbers of amino acids
257 contained in the four CN are summarized in Table S3. We can calculate that the four CN contain a
258 total of 784 amino acids, of which only 27 are phenylalanine. The resulting CN structure has a large
259 proportion of peptide bonds but a slight amount of benzene rings provided by phenylalanine.
260 Although the calculation is not accurate enough considering the different proportions of the four
261 monomers in CN, we can qualitatively analyze the proportion of the benzene rings and the peptide
262 bonds. Such a structure facilitates the tight binding of CN on the surface of graphene sheets and

263 forms graphene nanosheets decorated with CN during ball-milling, i.e. GN-CN.



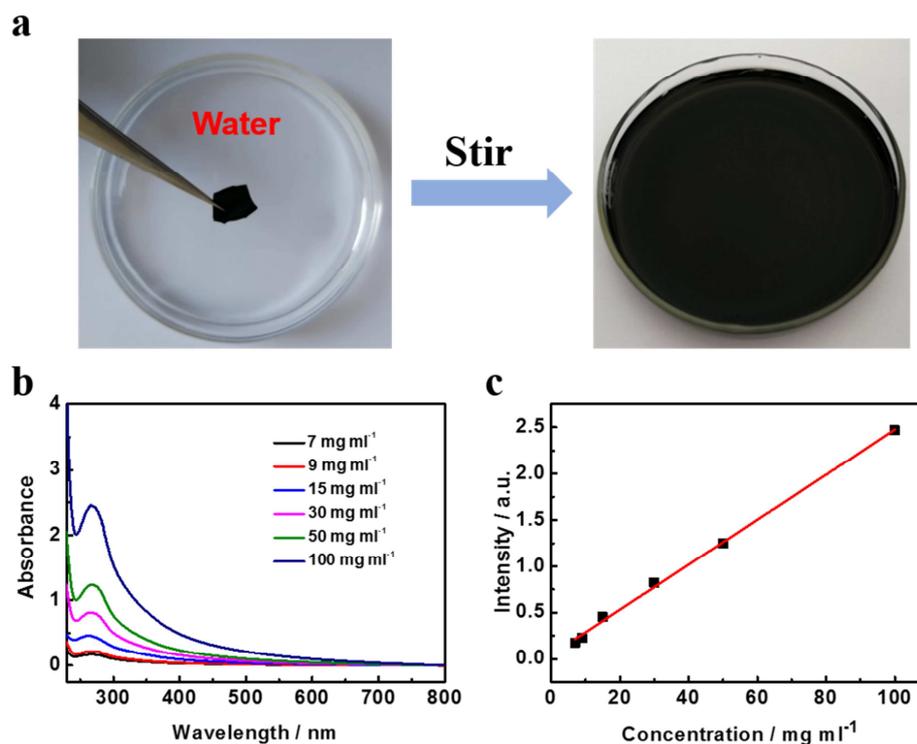
264 **Figure 3.** (a) Photographs of GN-CN dispersions with different pH from 1.0 to 13.0. (b) Zeta
 265 potential of GN-CN dispersions with a concentration of 1 mg ml^{-1} as a function of pH values. (c)
 266 Schematic illustration of GN-CN nanosheets' pH-responsive behavior.

267

268 3.2. Zwitterionic character of GN-CN

269 CN is an amphoteric protein with both amino and carboxyl groups on each end of the molecule
 270 chains. The special molecule structure endows it with an IP at around pH 4.6, at which the CN
 271 dispersion is unstable due to the absence of electrostatic repulsion. To determine whether the GN-CN
 272 exhibits the similar properties, we studied the stability of GN-CN dispersion at a series of pH values.
 273 Fig. 3a shows the stability of GN-CN dispersions at different pH values. It's obvious that GN-CN
 274 have aggregated at pH of near the IP of CN. To further understand the stability and surface charge of

275 GN-CN dispersion under different pH conditions, as shown in Fig. 3b, we studied the pH sensitivity
276 of zeta potential of GN-CN. When the pH is above the IP of CN, carboxyl ionization on CN
277 molecules makes GN-CN negatively charged, the electrostatic repulsion results in stable graphene
278 dispersions. We find that the GN-CN dispersion exhibits higher stability at pH=7-11 with zeta
279 potentials below -30 mV. Similarly, when the pH is below IP, graphene is positively charged due to
280 amino protonation, and graphene will aggregate due to the weak electrostatic repulsion when the pH
281 values is near the IP (pH=4-5). The change of zeta potential of GN-CN dispersion at acidic (pH=3)
282 and alkaline (pH=9) condition was also examined over 15 days (Fig. S9). The small change of zeta
283 potential and a negligible level of sediment in two months indicate that GN-CN can remain stable
284 over two months. The above analyses indicate that we have prepared a stable zwitterionic graphene
285 nanosheets with pH-responsive property.



286 **Figure 4.** (a) Schematic illustration of the cast-drying GN-CN film was redispersed in water to form
 287 a homogeneous graphene dispersion. (b) UV-vis spectra of GN-CN in water at different
 288 concentrations of 7, 9, 15, 30, 50, 100 mg ml⁻¹, respectively. (c) The plot of the absorbance intensity
 289 versus the concentration.

290

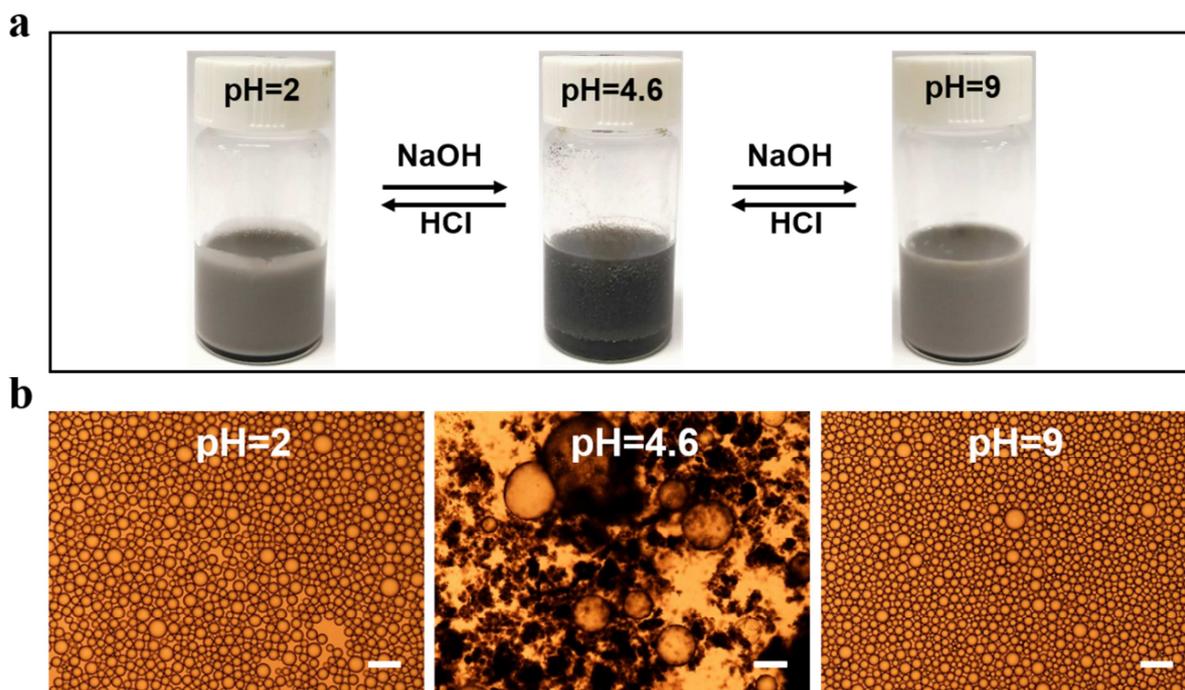
291 3.3. Redispersibility of GN-CN

292 The redispersibility is of great significance for graphene, because solid state graphene is more
 293 favorable to practical application due to its convenient storage and easy transportation³⁵. To evaluate
 294 the redispersibility of GN-CN, we dispersed the GN-CN film prepared by cast-drying into deionized
 295 water at room temperature. Interestingly, the GN-CN exhibits high re-dispersion concentration and
 296 fast dissolution rate simultaneously. As shown in Fig. 4a, a small piece of cast-drying GN-CN film
 297 was thrown into deionized water and stirred with a glass rod, then a homogeneous and stable GN-CN
 298 dispersion was formed. The redissolution process can be completed in a few minutes, which was

299 recorded by a video shown in Video. S1.

300 To ascertain the maximum concentration of GN-CN after re-dispersion, GN-CN dispersions with
301 different concentrations were prepared by adding a certain amount of cast-drying GN-CN to 3 ml
302 deionized water by sonication. To our surprise, the dispersion concentration of GN-CN was up to 100
303 mg ml^{-1} and corresponding concentration of graphene is 21.4 mg ml^{-1} . To the best of our knowledge,
304 it's the highest dispersible concentration for liquid-exfoliated graphene. As shown in Fig. 4b, UV-vis
305 spectra were measured for GN-CN dispersions with different concentration after redissolution. To
306 ensure the transmittance of the high concentrated graphene, all specimens were diluted with
307 deionized water to a factor of 100 before the test. The plot of the absorbance intensity at 264 nm as a
308 function of GN-CN concentration was shown in Fig. 4c. The plot shows a good linear relationship
309 when the GN-CN concentrations were below 100 mg ml^{-1} . We found that when the GN-CN
310 concentration was above 100 mg ml^{-1} , the complete dissolution of GN-CN became difficult even
311 with long sonication time. The re-dispersion property of GN-CN is attributed to the excellent
312 solubility of CN in water at alkaline condition and its tight binding on graphene surface. When
313 GN-CN was dispersed in water, CN dissolved and ionized rapidly and overcomes the π - π interaction
314 between graphene sheets by electrostatic repulsion, resulting in a stable GN-CN dispersion. We tried
315 to disperse the cast-drying GN-CN in different organic solvents by sonication but failed due to the
316 insolubility of CN in organic solvents. The result demonstrates the important role of CN in the
317 redispersibility of GN-CN.

318



319 **Figure 5.** (a) Toluene-in-water emulsions prepared at different pH values using 9 mg ml^{-1} GN-CN
320 dispersion. The volume ratio of toluene to water is 2:1. (b) Optical microscopy images of emulsions
321 at the corresponding pH values. All scale bars are $100 \mu\text{m}$.

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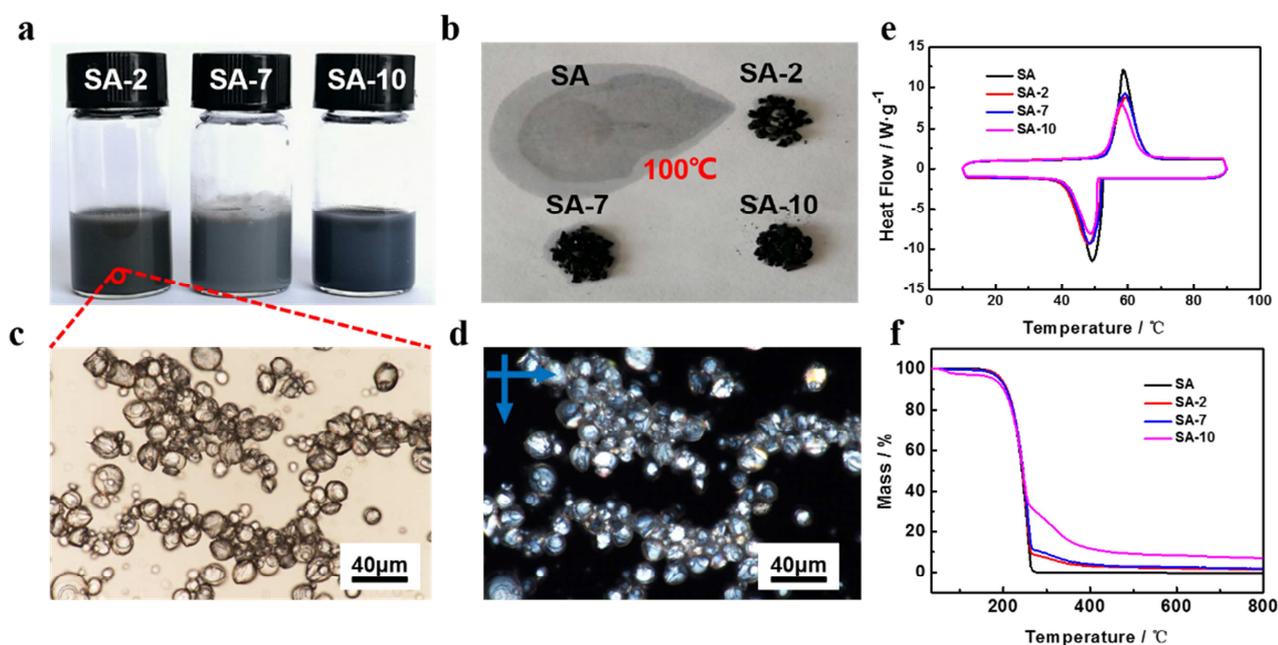
323 3.4. Pickering emulsion stabilized by GN-CN

324 As we all know, Pickering emulsion is a solid colloidal emulsion stabilized by solid particles with
325 moderate hydrophilicity and hydrophobicity. There are a few reports on the double electric emulsifier
326 stabilizing Pickering emulsions^{36, 37}. In order to study the potential of the zwitterionic GN-CN as
327 Pickering emulsifier, the mixtures of toluene and GN-CN dispersions with different pH values were
328 emulsified with a homogenizer. Toluene was used here because the π - π interaction between the
329 benzene ring and graphene may facilitate producing stable emulsions with uniform size. Fig. 5a
330 shows the effect of pH of GN-CN dispersions on emulsion stability. Stable emulsions are obtained

331 under both acidic (pH=2) and alkaline (pH=9) conditions and the spherical droplets are uniform in
332 size with a diameter far less than 100 μm shown in Fig. 5b. However, the weak electrostatic
333 interaction of GN-CN leads to strong hydrophobicity at pH near the IP of CN, which increases its
334 affinity to oil phase. As a result, the GN-CN aggregates enter the toluene phase of the upper layer
335 and few emulsion droplets with uniform size of several hundreds of micrometers were observed in
336 Fig. 5b. The optical micrographs of emulsions over a wide pH range are shown in Fig. S10,
337 indicating stable Pickering emulsions are obtained over a wide pH range except near the IP (pH
338 4-4.6). Furthermore, the size distributions of emulsion droplets were counted shown in Fig. S11. The
339 result shows that more than 90% droplets of emulsions prepared at different pH values distribute
340 between 10 and 50 μm in size. The emulsions prepared at pH=4 and 4.6 are not counted because it's
341 hard to find enough droplets.

342 In order to explain the excellent emulsification effect of GN-CN, CN was used to emulsify toluene
343 as a control under the same conditions with GN-CN. The optical microscopy images in Fig. S12
344 shows the similar emulsification effect of CN with GN-CN. We speculate the excellent emulsifying
345 property may be attributed to the amphiphilic structure of CN and its tight binding to the graphene
346 surface, which endows graphene a similar amphiphilic character. Such an amphiphilic GN-CN is
347 very promising in emulsions field such as foods, cosmetics etc. Thus, it is necessary to evaluate
348 biocompatibility of GN-CN by cytotoxicity test. The cytotoxicity was evaluated by culturing MLg
349 cells with GN-CN dispersions for 24 h. The live/dead cell staining assay shows excellent cell
350 viability and it's difficult to find significant dead cells (Fig. S13A). Furthermore, the CCK-8 assay
351 result shows high cell viability varies in a narrow range of 93.6%-96.2% (Fig. S13B). The cell
352 viability remained >93% even at a GN-CN concentration of 40 $\mu\text{g ml}^{-1}$, which is higher than that

353 culture with graphene oxide and other graphene exfoliated by exfoliating agents^{38, 39}. The negligible
 354 cytotoxicity indicates that GN-CN has full potential for the above mentioned emulsion applications.
 355



356 **Figure 6.** (a) Photographs of mPCMs emulsions emulsified at different pH values after cooling
 357 naturally at ambient temperature for 24h. (b) Photographs of SA and different mPCMs on a hot plate
 358 set 100°C. (c) Optical microscopy images of SA-2 emulsion. (d) Optical microscopy images of (c)
 359 after adding polarizer (e) DSC thermograms of SA and mPCMs. (f) TGA thermograms of SA and
 360 mPCMs.

361

362 3.5. Microcapsule phase change materials (mPCMs) prepared by GN-CN

363 Microcapsule phase change materials were considered promising thermal energy storage materials
 364 and possess high latent heat, high heat-transfer area and efficiency and exceptional shape stability
 365 due to unique core-shell structure^{40, 41}. Graphene has been used as Pickering emulsifier to prepare
 366 mPCMs due to the excellent electronic, thermal properties and large surface area⁴⁰⁻⁴³. However, most

367 of graphene emulsifiers are obtained by physical decoration or chemical modification which may
368 involve complex preparation process and toxic reagents. Considering facile preparation and
369 environmental-friendly characteristics of GN-CN, we studied the potential of preparing mPCMs
370 using GN-CN as Pickering emulsifier. To obtain mPCMs, the mixture of SA and GN-CN dispersion
371 with different pH values was emulsified with a homogenizer at 90 °C.

372 Fig. 6a shows the appearance of mPCMs emulsion prepared with different GN-CN dispersions.
373 For example. SA-2 is the mPCMs prepared with GN-CN dispersion of pH 2. It can be obviously
374 observed the difference in colour of the three mPCMs emulsions. We put a drop of the SA-2
375 emulsion onto a slide and let it dry at room temperature for a while, then the morphology of SA-2
376 was observed by optical microscope shown in Fig. 6c. The SA capsules are spherical and the size is
377 about 10-20 μm . Owing to the evaporation of water during the drying process, the SA capsules tend
378 to come together. To illustrate the capsule is stearic acid microsphere, optical microscopy images of
379 SA-2 after adding polarizer was observed as shown in Fig. 6d, the light sphere represents crystalline
380 structure of stearic acid. Moreover, the microsphere is surrounded by a membrane-like structures,
381 suggesting that SA was encapsulated by GN-CN and formed the core-shell structure. For comparison,
382 the SA-7 and SA-10 exhibit irregular morphology shown in Fig. S14a, c. Interestingly, SA-10
383 exhibits a rod-like liquid crystal structure shown in Fig. S14d. The reason may be that sodium
384 stearate was formed by the reaction of molten SA and sodium hydroxide under alkaline conditions at
385 90 °C, then the interaction between sodium stearate and GN-CN leads to the sodium stearate rod
386 structure coated with GN-CN. The morphology of mPCMs was further confirmed by SEM shown in
387 Fig. S15.

388 **Table 1.** Thermal properties of SA and mPCMs

	$\Delta H_m / \text{J g}^{-1}$	T_m / \square	$\Delta H_c / \text{J g}^{-1}$	T_c / \square
SA	196.2	58.63	194.5	49.27
SA-2	179.7	59.54	173.9	48.17
SA-7	174.5	59.15	167.8	48.54
SA-10	146.7	57.93	129.9	48.74

389

390 To investigate the thermal performance of mPCMs, the pH values of the emulsions were adjusted
391 near the IP of CN. Then the sediments were collected by filtration and drying. Fig. 6b shows the
392 photograph of SA and mPCMs on a hot plate at 100 °C. It's clear that SA has completely melted and
393 the paper is soaked. SA-7 exhibits an obvious leakage, suggesting the melted SA can't be trapped
394 effectively by GN-CN during the phase change process. The irregular core-shell structure of SA-7
395 also explains this phenomenon. However, SA-2 exhibits an excellent shape stability due to the
396 effective trapping of melting SA cores, which prevents the melted SA from leakage. It was strange
397 that SA-10 also didn't show significant leakage. That may have a lot to do with its liquid crystal
398 structure and the formation mechanism of SA-10 need to be further explored and discussed. In
399 addition, high latent heat is of great significance to mPCMs because it determines the energy storage
400 potential. Therefore, DSC thermograms of mPCMs were measured as shown in Fig. 6e. The
401 corresponding thermal properties data of SA and mPCMs were listed in Table 1. The heat of melting
402 (ΔH_m) and the heat of crystallization (ΔH_c) of mPCMs are both lower than neat SA, indicating the
403 existence of GN-CN component. Certainly, a decrease of GN-CN component will increase latent
404 heat but at the expense of shape stability. SA-2, SA-7 and SA-10 exhibit ΔH_m of 91.6%, 88.9% and
405 74.8% of neat SA respectively, indicating SA-2 possesses the highest latent. Meanwhile, the TGA

406 profile shows the highest weight loss occurs to SA-2, indicating the thinnest shell. It should be noted
407 that the significant decrease of latent heat of SA-10 may be attributed to graphene's inhibition of SA
408 crystallization due to the dispersion of graphene nanosheets in SA matrix⁴⁴. We can conclude that an
409 mPCMs with high latent heat and regular core-shell structure can be prepared at pH 2 using GN-CN
410 as Pickering emulsifier.

411

412 **4. Conclusions**

413 In summary, we have demonstrated a green, facile and large-scale route to produce few layered
414 graphene nanosheets decorated and stabilized by CN using a ball milling method. Benefiting from
415 the strong hydrophobic interaction, CN can be used effective exfoliating and stabilizing agent. The
416 resultant GN-CN shows exceptional redispersibility and dispersible concentration can be up to 100
417 mg ml⁻¹ and even higher. Meanwhile, the aqueous GN-CN dispersion is stable under various pH
418 values except near the IP of CN due to the weak electrostatic repulsion. Furthermore, the amphiphilic
419 structure gives GN-CN excellent emulsifying property and it can be used as a good Pickering
420 emulsion stabilizer over a wide pH values range, the negligible cytotoxicity makes GN-CN a
421 potential candidate for emulsion applications in food and cosmetic fields. To explore the potential
422 application of the GN-CN, SA/graphene composite microcapsule phase change material was
423 successfully prepared at pH 2 using a Pickering emulsion technology, which GN-CN serves as an
424 emulsion stabilizer. We presume that the novel graphene nanomaterial also can be used to prepare
425 graphene/casein composite gel by cross-linking CN proteins through our preliminary attempt, or as
426 an optical or electronic sensor.

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430 **Appendix A. Supplementary data**

431 Supplementary data associated with this article can be found, in the online version, at
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Declaration of interests

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The authors declare the following financial interests/personal relationships which may be considered as potential competing interests: